

# Synthesis and Crystal Structure of Orthorhombic $\text{NaSn}_2\text{Cl}_5$ : A New Type of $\text{AB}_2\text{X}_5$ Compound

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DEDICATED TO PROFESSOR HANS UWE SCHUSTER† ON THE OCCASION OF HIS 65TH BIRTHDAY

The phase diagram of the quasi-binary system  $\text{NaCl-SnCl}_2$  has been derived from DSC and X-ray investigations. In this system a dystectic  $\text{NaSn}_2\text{Cl}_5$  compound phase was found, which was obtained by grinding mixtures of  $\text{NaCl}$  and  $\text{SnCl}_2$  in a closed ball mill and then subsequent tempering at temperatures about  $170^\circ\text{C}$  in evacuated sealed silica ampoules. The single-crystal structure of  $\text{NaSn}_2\text{Cl}_5$  was determined. It crystallizes with a new orthorhombic-type arrangement [space group  $Pnmm$ ;  $a \approx 818.09(5)$ ,  $b = 1203.1(1)$ ,  $c = 871.63(8)$  pm;  $Z = 4$ ]. The structure was solved by Patterson methods with the MolEN program, and refined to a final  $R$  value of 0.019 for 717 unique reflections. The sodium ions are located in almost octahedral voids of Cl ions. Two kinds of tin ions manifest themselves structurally by a great distortion of their coordination environment, three Cl ions are situated at  $<272.77$  pm from the tin ion, which is a reasonable Sn-Cl bond length, but the other five or six are repelled by the lone pair to distances of  $>306.75$  pm. Different from the structure of  $\text{U}_3\text{Se}_5$ , it is another distorted  $\text{Rh}_5\text{Ge}_3$  structure. In this paper the structural relationship between  $\text{NH}_4\text{Pb}_2\text{Cl}_5$  and  $\text{NaSn}_2\text{Cl}_5$  is elucidated systematically by means of the group theoretical representation of group-subgroup relations. © 1995 Academic Press, Inc.

## INTRODUCTION

Ternary halides of the formula  $\text{AB}_2\text{X}_5$  ( $A = \text{K, In, Tl, NH}_4$ ;  $B = \text{Sr, Sn, Pb}$ ;  $X = \text{Cl, Br, I}$ ) have been the object of investigation for a long time, and early work dates back to the beginning of this century. They were often prepared from aqueous solutions containing the binary compounds in appropriate concentrations. In the course of thermoanalytical studies of quasi-binary systems  $\text{AX-BX}_2$ , many of these compounds were discovered as dystectic or peritectic points in the phase diagrams, but there were contradictory findings abounding in the literature.

In 1937, Powell and Tasker (1) reported on the structure of  $\text{NH}_4\text{Pb}_2\text{Br}_5$ , they had prepared from aqueous solution. Later an orthorhombic polymorph was described,

and in 1968 Jansen (2) discovered a whole series of such "orthorhombic" compounds. This investigator, however, prepared these compounds by solid-state reactions. It was not until 1976 that Keller (3) and independently Ras *et al.* (4) elucidated the structure of one of the hitherto "orthorhombic" representatives, viz., that of  $\text{NH}_4\text{Pb}_2\text{Cl}_5$ , showing that it was monoclinic with almost orthogonal metrics.

In the 1980s, Beck *et al.* (5–7) studied systematically several series of  $\text{AB}_2\text{X}_5$  compounds containing  $ns^2$  cations with an inert or lone electron pair in the  $A$  or  $B$  position, in both or in one of them. They pointed out that all known  $\text{AB}_2\text{X}_5$  compounds of this family can clearly be separated into the monoclinic  $\text{NH}_4\text{Pb}_2\text{Cl}_5$  type and the tetragonal  $\text{NH}_4\text{Pb}_2\text{Br}_5$  type on the basis of the size relations  $A/X$  and  $B/X$ . The former is stabilized by lower  $A/X$  and higher  $B/X$  values, the latter is stabilized by larger  $A$  and smaller  $B$  ions.

The  $\text{NaCl-SnCl}_2$  system has not been described until now, although in 1913 Rack (8) and in 1988 Beck and Nau (9) reported on it. Due to synthetic difficulties, a ternary compound was not found. In this paper, one way to yield such a compound in a more convenient way is introduced. Orthorhombic  $\text{NaSn}_2\text{Cl}_5$ , a very distorted  $\text{Rh}_5\text{Ge}_3$ -type new structure, is reported for the first time.

## EXPERIMENTAL

### 1. Preparation of the Starting Materials

In order to avoid the formation of hydrates and other by-products, all starting materials and the products were synthesized and handled under dry argon gas. The  $\text{NaCl}$  used (Merck p.a. grade) was dried under vacuum at temperatures up to  $300^\circ\text{C}$  for about 3 hr. The preparation of anhydrous stannous chloride (10) was achieved by treatment of the hydrate  $\text{SnCl}_2 \cdot 2\text{H}_2\text{O}$  with acetic anhydride. The dehydration was almost instantaneous, much heat was evolved. The anhydrous salt separated, and after being washed free from acetic acid with dry ether it was dried at temperature  $110^\circ\text{C}$  for several hours.

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## 2. Polycrystalline Samples

Polycrystalline samples of ternary halides in the  $\text{NaCl}$ – $\text{SnCl}_2$  system were synthesized by grinding appropriate mixtures of  $\text{NaCl}$  and  $\text{SnCl}_2$  in closed ball mills for several hours, subsequently pressing them together under about 5 tons, and then annealing for 4 weeks at temperatures about  $170^\circ\text{C}$  in evacuated sealed quartz ampoules.

All starting materials and products were characterized by X-ray analysis using the Guinier-600 camera technique with Quartz as internal standard ( $\alpha\text{-SiO}_2$ ,  $a = 491.36$  pm and  $c = 540.54$  pm), using  $\text{CuK}\alpha_1$  radiation. The unit-cell dimensions of  $\text{NaSn}_2\text{Cl}_5$  were calculated by means of least-squares methods (LSUCR) (11). High-temperature X-ray diffraction patterns were obtained with an Enraf–Nonius Guinier Simon FR533 camera, using  $\text{CuK}\alpha_1$  radiation. The samples were sealed in quartz capillaries taken for sample holders. The heating and cooling rates used were  $10^\circ\text{C/hr}$  and the running rate of the film was  $1.5$  mm/hr. The experimental density was measured pycnometrically under an argon gas box.

Differential scanning calorimetry (DSC) measurements were performed with a Perkin–Elmer DSC7 calorimeter, with samples ( $\sim 5$  mg) sealed in closed gold crucibles. An empty sample pan was used as a reference. The heating rates were  $5 \sim 10^\circ\text{C/min}$ . For temperature calibration, the melting points and phase transitions of  $\text{KNO}_3$  and  $\text{AgCl}$  were used.

Raman spectra, with samples taken in sealed glass capillary tubes, were measured on a Dilor OMARS 89 multi-channel Raman spectrograph with the usual right-angle geometry (spectral slit width  $< 4$   $\text{cm}^{-1}$ ). For excitation, the  $514.5\text{-nm}$  line of an  $\text{Ar}^+$  ion laser was employed (laser power at the sample was about  $150$  mW). The integration time was  $1\text{--}30$  sec; the number of accumulations were  $30\text{--}50$ . Infrared spectra were recorded on a Bruker IFS 113  $\nu$  Fourier transform interferometer (resolution  $< 4$   $\text{cm}^{-1}$ ), using Nuyol mull as matrices.

## 3. Single Crystals

Colorless crystals of  $\text{NaSn}_2\text{Cl}_5$  were grown by fusing a stoichiometric mixture of  $\text{NaCl}$  and  $\text{SnCl}_2$  and tempering at temperatures about  $165^\circ\text{C}$  for about several weeks.

A crystal with well developed faces of the approximate size  $0.12 \times 0.2 \times 0.25$  mm was mounted in a sealed glass capillary. The space group was inferred from precession photographs of  $hk0$ ,  $hk1$ ,  $hk2$ ,  $h0l$ ,  $h1l$ , and  $h2l$  layers with a Buerger precession camera (Firma Huber, Rimsting). Intensity data were collected to  $2\theta < 49.8^\circ$  on an Enraf–Nonius CAD4 diffractometer using graphite-monochromatized  $\text{MoK}\alpha$  radiation ( $\lambda = 0.71069$  Å) (12). An orientation matrix for data collection were obtained from least-squares refinements, using the setting angles of 25 reflections in the range  $16^\circ < \theta < 24^\circ$ , measured by the

computer controlled diagonal slit method of centering. The intensities of three standard reflections were checked every 166 min, and orientation control was performed every 200 measured reflections. The data were corrected for Lorentz and polarization effects. Empirical absorption corrections from the program Enraf–Nonius were applied to the data. The structure of  $\text{NaSn}_2\text{Cl}_5$  was solved by using the program MolEN (13).

## RESULTS AND DISCUSSION

### 1. The $\text{NaCl}$ – $\text{SnCl}_2$ System

On the basis of the data obtained from DSC and X-ray phase analyses the equilibrium diagram of the  $\text{NaCl}$ – $\text{SnCl}_2$  system is given in Fig. 1. Tin(II) chloride forms one compound with sodium chloride  $\text{NaSn}_2\text{Cl}_5$ . The compound melts congruently at  $205^\circ\text{C}$ , and there are no polymorphic transformations up to the melting point.

### 2. Synthesis of Monophase Samples of $\text{NaSn}_2\text{Cl}_5$

Because there are large differences in the melting points between  $\text{NaCl}$  ( $800^\circ\text{C}$ ) and  $\text{SnCl}_2$  ( $240^\circ\text{C}$ ) and the reactivity of the products formed is quite slow, mixtures of the reactants and products always were obtained by normal ceramic or fusing methods. In order to speed up reaction rates, in the solid-state system, normally it is recommended to raise the temperature as high as possible. In this system, however, additional problems occur at high temperatures, e.g., the volatilization of  $\text{SnCl}_2$ . So

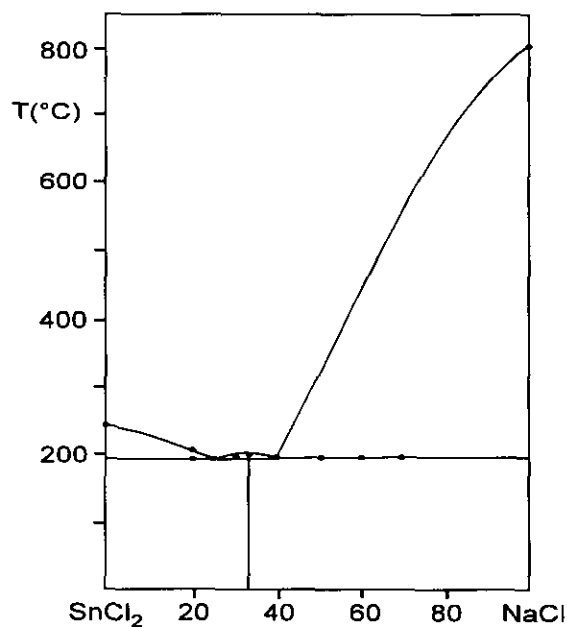


FIG. 1. Phase diagram of the  $\text{NaCl}$ – $\text{SnCl}_2$  system.

the reaction takes place only at  $\sim 170^\circ\text{C}$ , relatively, a low temperature. On the basis of this view, we have synthesized single-phase  $\text{NaSn}_2\text{Cl}_5$  by means of grinding mill and long-time tempering techniques.

### 3. Crystal Structure

Precession photographs of  $\text{NaSn}_2\text{Cl}_5$  display  $\text{mmm}$  symmetry with extinctions ( $0kl$ , with  $k + l = 2n$ ,  $h0l$  with  $h + l = 2n$ , and  $h00$ ,  $0k0$ , and  $00l$  with  $h = 2n$ ,  $k = 2n$ , and  $l = 2n$  respectively), which are characteristic of space groups  $Pn\bar{m}$  and  $Pnn2$ . The centrosymmetric group was found to be correct during the structure refinement.

The structure was solved using the Patterson heavy-atom method, which revealed the positions of two atoms. The remaining atoms were located in succeeding difference Fourier syntheses. The structure was refined with a full-matrix least-squares treatment, where the function minimized was  $w(|F_o| - |F_c|)^2$  and the weight  $w$  is defined at  $1/\sigma^2(F)$  for all observed reflections. Anomalous dispersion effects were included in  $F_c$ , the values for  $f'$  and  $f''$  were taken from Cromer and Waber (14). Only the 649 reflections with  $[|F_o| \geq 2.0\sigma(F_o)]$  were used in the refinements. The final cycle of refinement included 45 variable parameters and converged (largest parameter shift was 0.01 times is esd) to  $R$  and  $R_w$  is 0.019 and 0.026, respec-

TABLE 1  
Crystal Data, Data-Collection Parameters, and Details of  
Structure Refinements for  $\text{NaSn}_2\text{Cl}_5$

Space group	$Pn\bar{m}$ (No. 58)
Lattice constants <sup>a</sup> (pm)	$a = 818.09(5)$ $b = 1203.1(1)$ $c = 871.63(8)$
Unit cell volume (pm <sup>3</sup> )	$V = 857.93(9) \times 10^6$ $Z = 4$
Calculated density (g cm <sup>-3</sup> )	$d_{\text{th}} = 3.39$
Crystal dimensions (mm)	$0.12 \times 0.2 \times 0.25$
Temperature measured	293 K
Scan mode	$\omega/2\theta$ -scan
Measured region $2\theta_{\text{max}}$	$49.8^\circ$
$h, k, l$ ranges	$0 \leq h \leq 9, -14 \leq k \leq 14, 0 \leq l \leq 10$
No. of reflections measured	1389
No. of observed reflections $F_o > 2.0\sigma F_o$	649
No. of unique reflections	717
Parameters refined	45
Linear absorption coefficient	$\mu_{\text{MoK}\alpha} = 73.82 \text{ cm}^{-1}$
Extinction coefficient	$1.23(1) \times 10^{-6}$
Unweighted agreement factor	$R = 0.019$
Weighted agreement factor	$R_w = 0.026$
Weighting scheme	$w = 1/\sigma^2(F)$ $s = 1.951$
$\Delta\rho_{\text{max}}$ (e pm <sup>-3</sup> )	$0.670 \times 10^{-6}$
$\Delta\rho_{\text{min}}$ (e pm <sup>-3</sup> )	$-0.830 \times 10^{-6}$
Computer software	MolEN (Enraf-Nonius)

<sup>a</sup> Obtained from Guinier photograph.

TABLE 2  
Fractional Atomic Coordinates and Isotropic Thermal  
Parameters of  $\text{NaSn}_2\text{Cl}_5$

Atom	Site	$x$	$y$	$z$	$B_{\text{iso}}$
Na	4f	0.0	0.5	0.2527(1)	2.85(4)
Sn(1)	4g	0.04135(3)	0.16892(2)	0.0	2.014(5)
Sn(2)	4g	0.54279(3)	0.32717(2)	0.0	1.948(5)
Cl(1)	8h	0.17276(8)	0.29129(6)	0.28304(7)	2.31(1)
Cl(2)	4g	0.1885(1)	0.56338(9)	0.0	2.73(2)
Cl(3)	4g	0.33654(9)	0.10193(8)	0.0	2.05(1)
Cl(4)	4e	0.0	0.0	0.2051(1)	2.23(2)

Note.  $B_{\text{eq}} = 4/3 \sum_i \sum_j \beta_{ij} a_i^* a_j^*$ .

tively. The final difference Fourier map showed no significant residual electron density. The results of the refinement procedures are given in Tables 1, 2, and 3. Table 4 contains a list of all interatomic distances and angles.

In the  $\text{NaSn}_2\text{Cl}_5$ -type structure, the Sn ions are found in strongly distorted trigonal prisms of Cl ions with three additional Cl ions capping this polyhedron. The tricapped trigonal prisms are joined along  $[001]$  by common triangular faces, containing two kinds of tin ions alternatively. Two such polyhedra share the edge; these double prisms are connected by sharing the "free" edges of other double prisms given a pseudo-hexagonal arrangement, such structural units are again interconnected within the  $(001)$  planes in same manner, which is shown in Fig. 2. Three Cl ions on one side of the tin are at  $\leq 272.77$  pm, which is a reasonable Sn-Cl bond length, but the other are repelled by the lone pair to distances of  $> 306.75$  pm. The  $\text{Na}^+$  ions lie along pseudo-hexagonal axes with different  $z$  parameters. Sodium is coordinated to six Cl atoms forming a slightly distorted octahedron with Na-Cl distances varying between 279.48(9) and 289.35(7) pm (Fig. 3). The distortion of the  $\text{NaCl}_6$  octahedra results from sharing edges between them and corners with trigonal  $\text{SnCl}_6$  prisms.

TABLE 3  
Anisotropic Thermal Parameters ( $U_{ij}/100 \text{ pm}^2$ ) of  $\text{NaSn}_2\text{Cl}_5$

Atom	$U_{11}$	$U_{22}$	$U_{33}$	$U_{12}$	$U_{13}$	$U_{23}$
Na	5.3(1)	2.8(1)	2.66(9)	-0.2(1)	0.0	0.0
Sn(1)	2.17(1)	2.29(1)	2.19(1)	0.19(1)	0.0	0.0
Sn(2)	2.32(1)	2.61(1)	2.47(1)	0.05(1)	0.0	0.0
Cl(1)	3.36(3)	2.72(4)	2.70(2)	0.12(3)	0.62(3)	0.31(3)
Cl(2)	3.36(4)	4.32(5)	2.68(4)	-0.162(4)	0.0	0.0
Cl(3)	1.91(3)	3.11(4)	2.79(4)	0.14(4)	0.0	0.0
Cl(4)	3.90(4)	2.63(5)	1.94(3)	-0.45(4)	0.0	0.0

Note.  $U_{ij}$  is defined as  $T = \exp(-2\pi^2 \sum_i \sum_j U_{ij} h_i h_j a_i^* a_j^*)$ .

TABLE 4  
Interatomic Distances (pm) and Angles (°) of NaSn<sub>2</sub>Cl<sub>5</sub>

Na <sup>0</sup> -Cl(1) <sup>0,1</sup>	289.35(7)	Cl(1) <sup>0</sup> -Na <sup>0</sup> -Cl(1) <sup>1</sup>	169.51(4)
Na <sup>0</sup> -Cl(2) <sup>0,2</sup>	279.48(9)	Cl(1) <sup>0,1</sup> -Na <sup>0</sup> -Cl(2) <sup>0,2</sup>	92.25(3)
Na <sup>0</sup> -Cl(3) <sup>3,4</sup>	281.75(9)	Cl(1) <sup>0,1</sup> -Na <sup>0</sup> -Cl(2) <sup>2,0</sup>	96.01(3)
		Cl(1) <sup>0,1</sup> -Na <sup>0</sup> -Cl(3) <sup>3,4</sup>	94.36(3)
Sn(1) <sup>0</sup> -Cl(3) <sup>0</sup>	254.59(8)	Cl(1) <sup>0,1</sup> -Na <sup>0</sup> -Cl(3) <sup>4,3</sup>	77.54(3)
Sn(1) <sup>0</sup> -Cl(4) <sup>0,8</sup>	272.77(6)	Cl(2) <sup>0</sup> -Na <sup>0</sup> -Cl(2) <sup>2</sup>	75.98(3)
Sn(1) <sup>0</sup> -Cl(1) <sup>0,5</sup>	306.75(7)	Cl(2) <sup>0,2</sup> -Na <sup>0</sup> -Cl(3) <sup>3,4</sup>	102.85(2)
Sn(1) <sup>0</sup> -Cl(1) <sup>6,7</sup>	359.14(7)	Cl(2) <sup>0,2</sup> -Na <sup>0</sup> -Cl(3) <sup>4,3</sup>	169.60(3)
Sn(1) <sup>0</sup> -Cl(2) <sup>2</sup>	372.9(1)	Cl(3) <sup>3</sup> -Na <sup>0</sup> -Cl(3) <sup>4</sup>	80.18(3)
Sn(2) <sup>0</sup> -Cl(2) <sup>9</sup>	256.25(9)	Cl(1) <sup>0</sup> -Sn(1) <sup>0</sup> -Cl(1) <sup>5</sup>	107.08(2)
Sn(2) <sup>0</sup> -Cl(1) <sup>10,11</sup>	259.58(7)	Cl(1) <sup>0,5</sup> -Sn(1) <sup>0</sup> -Cl(3) <sup>0,0</sup>	79.60(2)
Sn(2) <sup>0</sup> -Cl(3) <sup>0</sup>	319.2(1)	Cl(1) <sup>0,5</sup> -Sn(1) <sup>0</sup> -Cl(4) <sup>0,8</sup>	82.76(2)
Sn(2) <sup>0</sup> -Cl(4) <sup>12,10</sup>	332.46(7)	Cl(1) <sup>0,5</sup> -Sn(1) <sup>0</sup> -Cl(4) <sup>8,0</sup>	158.15(2)
Sn(2) <sup>0</sup> -Cl(1) <sup>0,5</sup>	392.89(7)	Cl(3) <sup>0,0</sup> -Sn(1) <sup>0</sup> -Cl(4) <sup>0,8</sup>	83.21(2)
Sn(2) <sup>0</sup> -Cl(2) <sup>0</sup>	405.9(1)	Cl(4) <sup>0</sup> -Sn(1) <sup>0</sup> -Cl(4) <sup>9</sup>	81.90(2)
		Cl(1) <sup>10</sup> -Sn(2) <sup>0</sup> -Cl(1) <sup>11</sup>	93.53(2)
		Cl(1) <sup>10,11</sup> -Sn(2) <sup>0</sup> -Cl(2) <sup>9,9</sup>	86.03(2)
		Cl(1) <sup>10,11</sup> -Sn(2) <sup>0</sup> -Cl(3) <sup>0,0</sup>	75.55(2)
		Cl(1) <sup>10,11</sup> -Sn(2) <sup>0</sup> -Cl(4) <sup>12,10</sup>	161.77(2)
		Cl(1) <sup>10,11</sup> -Sn(2) <sup>0</sup> -Cl(4) <sup>10,12</sup>	79.82(2)
		Cl(2) <sup>0</sup> -Sn(2) <sup>0</sup> -Cl(3) <sup>0</sup>	152.83(3)
		Cl(2) <sup>9,9</sup> -Sn(2) <sup>0</sup> -Cl(4) <sup>10,12</sup>	76.64(2)
		Cl(3) <sup>0</sup> -Sn(2) <sup>0</sup> -Cl(4) <sup>10,12</sup>	118.38(1)
		Cl(4) <sup>13</sup> -Sn(2) <sup>0</sup> -Cl(4) <sup>12</sup>	101.28(2)

Note. (0)  $x, y, z$ ; (1)  $-x, 1-y, z$ ; (2)  $-x, 1-y, -z$ ; (3)  $\frac{1}{2}-x, \frac{1}{2}+y, \frac{1}{2}+z$ ; (4)  $-\frac{1}{2}+x, \frac{1}{2}-y, \frac{1}{2}-z$ ; (5)  $x, y, -z$ ; (6)  $-\frac{1}{2}+x, \frac{1}{2}-y, \frac{1}{2}+z$ ; (7)  $-\frac{1}{2}+x, \frac{1}{2}-y, -\frac{1}{2}+z$ ; (8)  $-x, -y, -z$ ; (9)  $1-x, 1-y, -z$ ; (10)  $\frac{1}{2}+x, \frac{1}{2}-y, \frac{1}{2}-z$ ; (11)  $\frac{1}{2}+x, \frac{1}{2}-y, -\frac{1}{2}+z$ ; (12)  $\frac{1}{2}-x, \frac{1}{2}+y, -\frac{1}{2}+z$ .

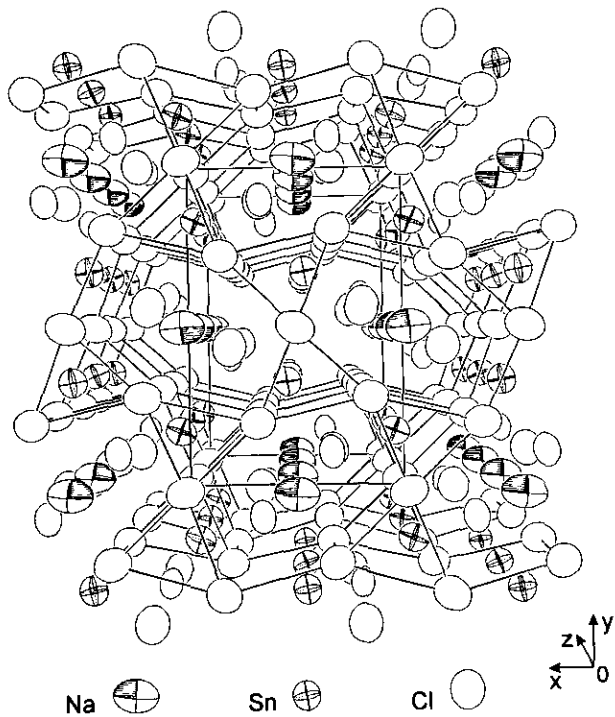


FIG. 2. Crystal structure of orthorhombic NaSn<sub>2</sub>Cl<sub>5</sub> (Ortep plot, 97% probability).

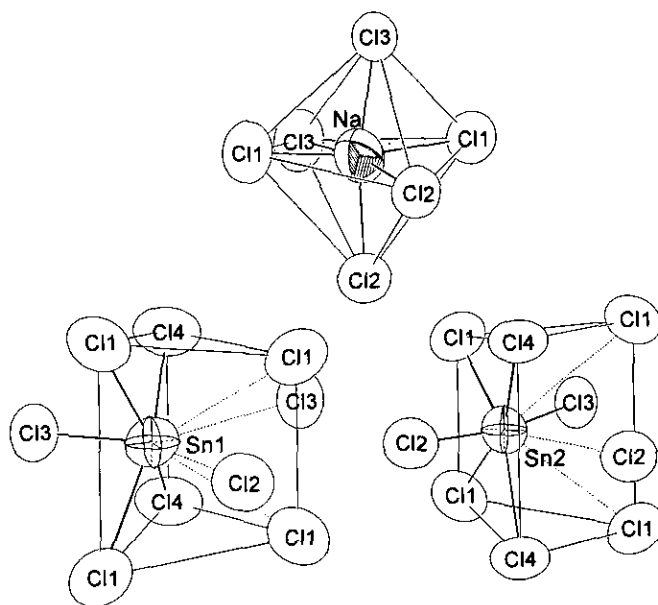


FIG. 3. Slightly distorted NaCl<sub>6</sub> octahedron (top) and strongly distorted SnCl<sub>5</sub> polyhedra (bottom) in the NaSn<sub>2</sub>Cl<sub>5</sub> structure.

#### 4. Raman and IR Spectra of NaSn<sub>2</sub>Cl<sub>5</sub>

Unit-cell group analysis ( $D_{2h}^{12}$ ) (15) yields the following distribution of the 96 vibrational modes:

$$\begin{aligned}\Gamma_{\text{Raman}} &= 13A_g + 13B_{1g} + 11B_{2g} + 11B_{3g} \\ \Gamma_{\text{infrared}} &= 8B_{1u} + 14B_{2u} + 14B_{3u} \\ \Gamma_{\text{acoustic}} &= B_{1u} + B_{2u} + B_{3u} \\ \Gamma_{\text{silent}} &= 9A_u.\end{aligned}$$

All the Raman bands are located in the 10–300 cm<sup>-1</sup> spectral region. The 108-K spectrum contains nine bands, viz., at 32, 57, 80, 88, 130, 143, 184, 251, and 265 cm<sup>-1</sup>. On cooling the samples, the spectrum became better resolved as the line widths narrowed (Fig. 4). The 79 cm<sup>-1</sup> band is split into two distinct components. In general, the peaks were found to shift to higher frequency on cooling, a behaviour consistent with the expected increase in force constant as the crystal contracted. For instance at 108 K, the 126 and 139 lines shift to 130 and 143 cm<sup>-1</sup>.

The IR spectra of NaSn<sub>2</sub>Cl<sub>5</sub> reveal up to eight bands, viz., at 48, 75, 122, 140, 153, 203, 246, and 303 cm<sup>-1</sup> (see Fig. 5).

#### 5. Structural Relationships

So far all AB<sub>2</sub>X<sub>5</sub> compounds (A = Na, K, In, Tl, NH<sub>4</sub>; B = Sr, Sn, Pb; X = Cl, Br, I) were found to belong to one of three structure types, viz., the NH<sub>4</sub>Pb<sub>2</sub>Br<sub>5</sub> type, space group  $I4/mcm$ ; the NH<sub>4</sub>Pb<sub>2</sub>Cl<sub>5</sub> type, space group  $P2_1/c$ ; and the NaSn<sub>2</sub>Cl<sub>5</sub> type, space group  $Pnmm$ . There

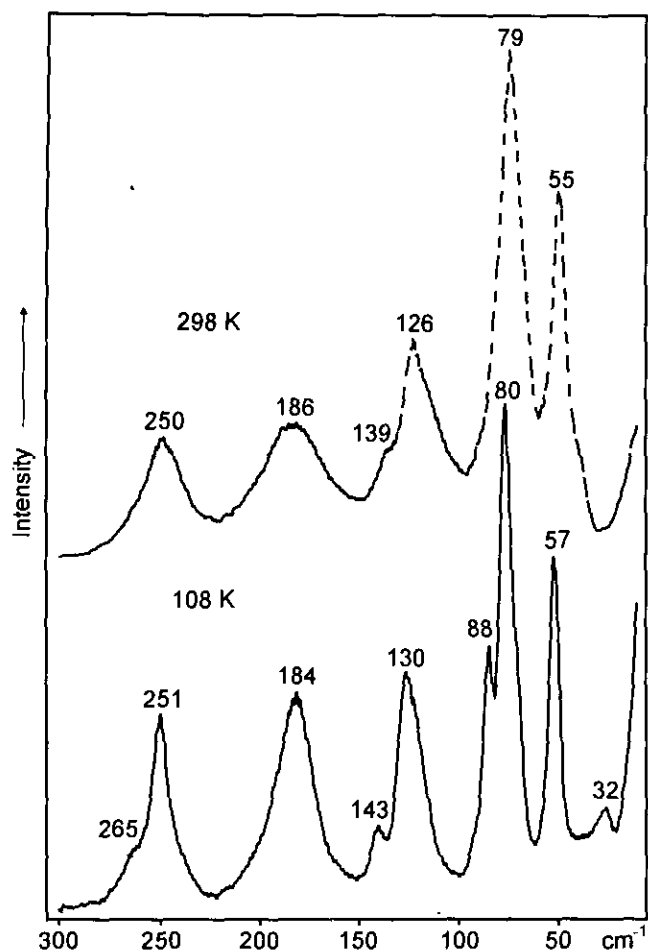


FIG. 4. Raman spectra of  $\text{NaSn}_2\text{Cl}_5$  recorded at ambient temperature (dashed line) and at liquid-nitrogen temperature (full line).

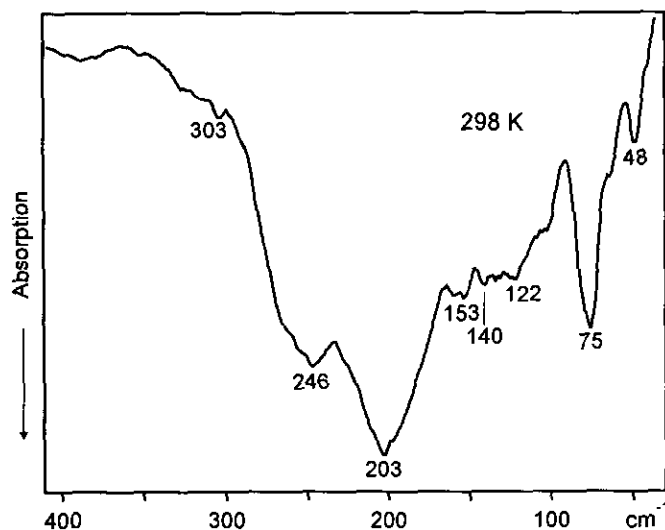


FIG. 5. IR spectrum of  $\text{NaSn}_2\text{Cl}_5$  at ambient temperature.

are close structural relationships between the  $\text{NH}_4\text{Pb}_2\text{Cl}_5$  and  $\text{NaSn}_2\text{Cl}_5$  types.

The following family tree (see Fig. 6) is an extension of such a diagram given earlier (16). Chemical formulas in parentheses indicate the content of the unit cell. Vacant sites are indicated by squares. Unconventional settings are used to demonstrate the symmetry relationships more clearly. The  $\text{Rh}_5\text{Ge}_3$  structure type plays a central role in this family of structures, where the different crystallographic sites of Ge are differentiated chemically.  $\text{La}_2\text{SnS}_5$  is another example for this structural arrangement.

The  $\text{U}_3\text{Se}_5$ -type structure derives from the  $\text{Rh}_5\text{Ge}_3$ - (anti-)type structure via small distortions, which results in doubling the  $c$  axis of  $\text{U}_3\text{Se}_5$  (or  $\text{Y}_2\text{HfS}_5$ ). No interchange of atomic positions is needed to go from one structural type to another (17, 18). In this branch, the  $\text{NH}_4\text{Pb}_2\text{Cl}_5$  structure finally results from an additional *translationgleich* reduction with loss of orthogonality, by which different cations are allowed on a formerly equivalent site. The  $\text{NaSn}_2\text{Cl}_5$ -type structure is another very distorted version of the  $\text{Rh}_5\text{Ge}_3$  structure by doubling the  $c$  axis *klassengleich*, as for  $\text{U}_3\text{Se}_5$ , but where different cations occupy a respective equivalent site.

The  $\text{Rh}_5\text{Ge}_3$ -type structure may be considered as a distorted deficient  $\text{Fe}_2\text{P}$  (=  $\text{Fe}_6\text{P}_3$ )-type structure, where  $\frac{1}{6}$  of the Fe positions are unoccupied (19). The  $\text{U}_3\text{Se}_5$  and

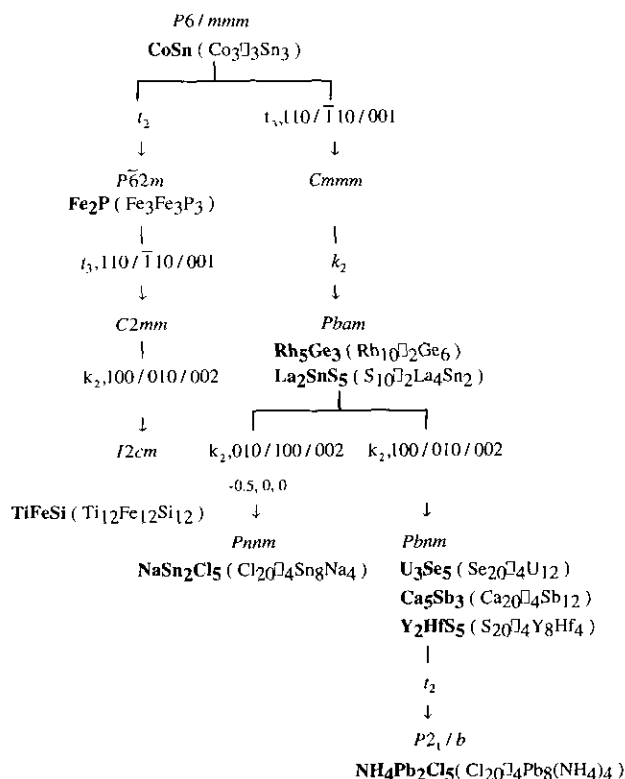


FIG. 6. Space-group relationships of "filled" CoSn-type structures.

$\text{NaSn}_2\text{Cl}_5$  (distorted  $\text{Rh}_5\text{Ge}_3$ -type) structures then correspond to the  $\text{TiFeSi}$  (distorted  $\text{Fe}_2\text{P}$ -type) structure (20). In terms of space-group relationships, however, the  $\text{Fe}_2\text{P}$ - and  $\text{Rh}_5\text{Ge}_3$ -type structures, as well as their distorted versions with lower symmetry and doubled pseudo-hexagonal axes ( $\text{TiFeSi}$ ,  $\text{U}_3\text{Se}_5$ ,  $\text{NH}_4\text{Pb}_2\text{Cl}_5$ , and  $\text{NaSn}_2\text{Cl}_5$ ), represent different subgroups of  $P6/mmm$ . This space group is represented by the  $\text{CoSn}$ -type structure, which may be considered as a higher symmetric

defect structure of both the  $\text{Fe}_2\text{P}$ - and the  $\text{Rh}_5\text{Ge}_3$ -type structure with  $\frac{1}{2}$  or  $\frac{1}{6}$  of the metal positions unoccupied, respectively.

Structure correlations by means of symmetry relations between space groups are to be of practical value. In order to elucidate more clearly the structure relationships between the  $\text{NaSn}_2\text{Cl}_5$ -type and the  $\text{NH}_4\text{Pb}_2\text{Cl}_5$ -type structures, the scheme of their crystal structures is shown in Fig. 7. For  $\text{La}_2\text{SnS}_5$ , two adjacent unit cells are

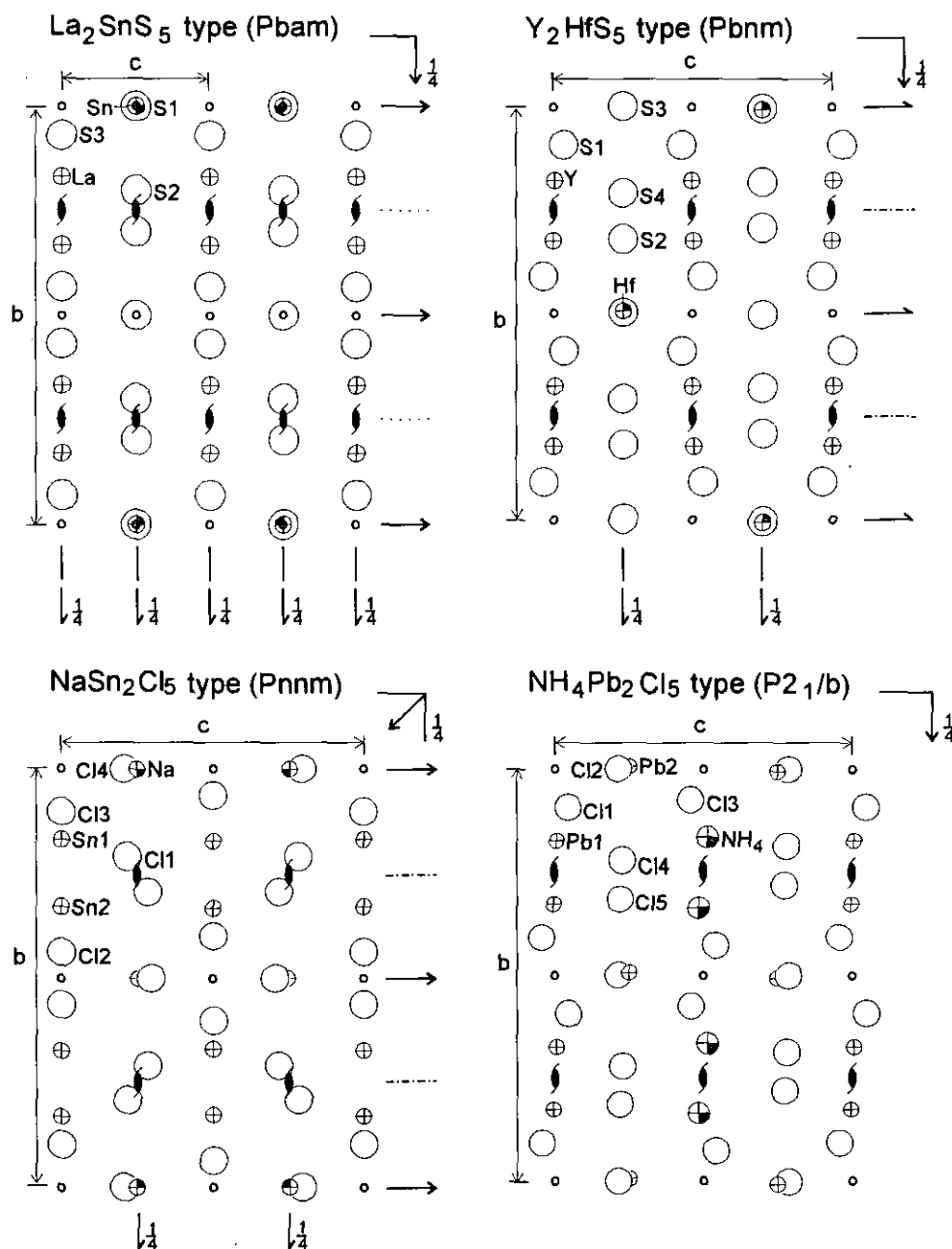


FIG. 7. Schematic representation of the crystal structures of  $\text{AB}_2\text{X}_5$  compounds. All symmetry elements of the respective space groups are included in order to show the way of thinning (*Auslichtung*) during the symmetry reduction.

included, and for  $\text{NaSn}_2\text{Cl}_5$ , the origin is moved to  $(-0.5, 0, 0)$ . All atoms lie on mirror planes perpendicular to the  $c$  axis. If one reduces the symmetry by index  $k_2$  leaving out half of the translations in the  $c$  direction, which is equivalent to doubling the  $c$  axis, only one-half of the mirror planes, one-half of the inversion centers, and one-half the screw axes can be retained.

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